



# Analytical Methods for Materials

## Laboratory Module #5

### Determination of Crystallite Size and Lattice Strain

#### Suggested Reading

- C. Suryanarayana and M.G. Norton, *X-ray Diffraction A Practical Approach*, (Plenum Press, New York, 1998), pages 207-221.
- B.D. Cullity and S.R. Stock, *Elements of X-ray Diffraction, 3<sup>rd</sup> Edition*, (Prentice-Hall, Upper Saddle River, NJ, 2001), Ch. 5, pages 167-176; and Ch. 14, pages 385-402.
- Y. Waseda, E. Matsubara, and K. Shinoda, *X-ray Diffraction Crystallography*, (Springer, New York, NY, 2011), Ch. 4, pages 123-126, 164-167.

# Objectives

- Upon completion of this module, the student will be able to determine the average crystal size and lattice strain in a specimen using X-ray peak broadening analyses.

# Background

- Bragg's law assumes that ideal conditions are maintained during diffraction.
  - The crystal is perfect
  - The X-ray beam is collimated and monochromatic
  - Etc...
- In practice, this condition never exists.

# Ideal Particle Size

- With powder XRD, the ideal particle size depends on the “relative” perfection of the polycrystalline sample.
  - Usually 0.5  $\mu\text{m}$  to 10  $\mu\text{m}$
- What are the implications of smaller crystals?
  - *The number of parallel planes (for XRD) is too low for a sharp diffraction maximum to build up. Thus, XRD peaks become “broadened”.*



# Crystal Size and Lattice Strain

- Measurements depend upon:
  - XRD peak locations
  - XRD peak widths
- Must take peak broadening into account
  - What causes peak broadening?

# Crystal Size and Lattice Strain – cont'd

- $d$ -spacings directly influence peak width.
- What can change  $d$ -spacing?
  - strains
  - compositional variations
  - instrumental variables
    - slits (filtering and focusing)
    - sample area
    - diffracting layer thickness
  - particle size

# More on the causes of broadening

- Instrument effects:
  - Imperfect focusing of diffracted beam
  - Unresolved  $\alpha_1$  and  $\alpha_2$  peaks
  - Finite widths of  $\alpha_1$  and  $\alpha_2$  peaks when resolved
- Crystal size
- Lattice strain

# Some Terminology

- Domain size:
  - Domains are the parts of an XRD specimen that diffract X-rays coherently. They form “substructure.” Domains are usually regions that are misoriented relative to one another (usually by  $<1^\circ$ ).
    - Subgrain boundaries, tilt boundaries, etc...
- Crystal (or crystallite) size:
  - Often used interchangeably with grain size. Refer to regions separated by large misorientation angles.
- Grain size:
  - Single crystal regions separated by large misorientation angles



# Corrections for instrumental broadening

We must first determine the individual contributions to peak broadening

$$\begin{aligned} &\text{Peak Broadening} \\ &\quad || \\ &\text{instrumental component} \\ &\quad + \\ &\text{strain/particle size} \end{aligned}$$

# Corrections for instrumental broadening

- Let  $\beta$  = peak width
- $\beta_{\text{observed}} = \beta_{\text{instrumental}} + \beta_{\text{strain and particle size}}$   
or
- $\beta_o = \beta_i + \beta_r$  ← *What we desire*
- $\beta_o$  is usually determined using well-annealed powders


## Corrections -- Cont'd

- Rewrite equation for lattice strain and particle size effects:

$$\beta_r = \beta_o - \beta_i$$

- This expression holds for a Lorentzian\* peak profile, however, many peaks have a Gaussian<sup>‡</sup> profile:

$$\beta_r^2 = \beta_o^2 - \beta_i^2$$

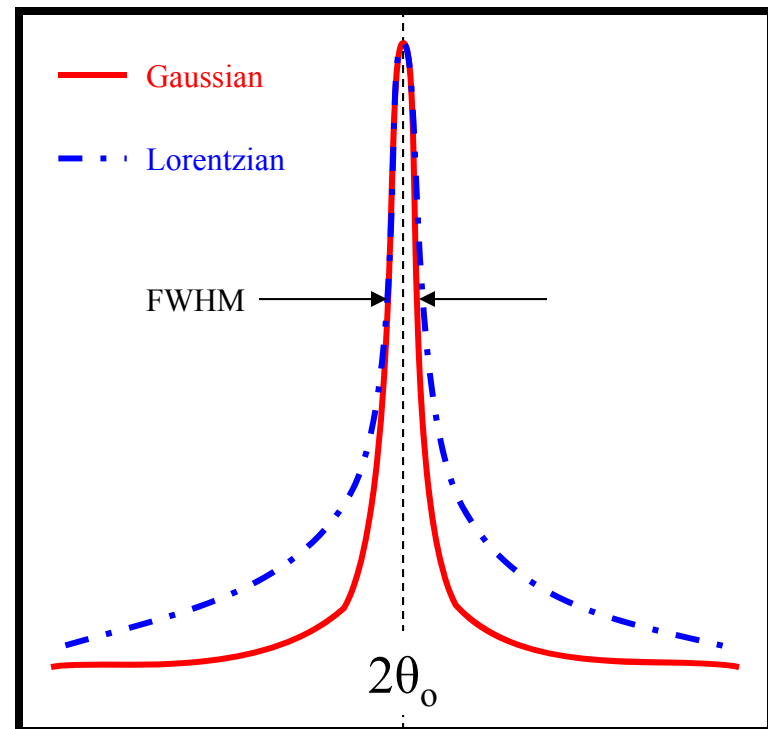
 *Most texts use this expression for simplicity*

- When in doubt, however, use:

$$\beta_r = \sqrt{(\beta_o - \beta_i) \sqrt{(\beta_o^2 - \beta_i^2)}}$$

# Corrections -- Cont'd

- Lorentzian\* peak profile:
  - The peak falls away slowly as  $(2\theta - 2\theta_0)^{-2}$  from the center of the peak ( $2\theta_0$ )
- Gaussian<sup>‡</sup> peak profile:
  - The peak is “sharper.”
  - Actually there is more to it than this.
- Modern XRD systems generally include software to assist you in fitting peaks.



# Broadening due to crystallite Size

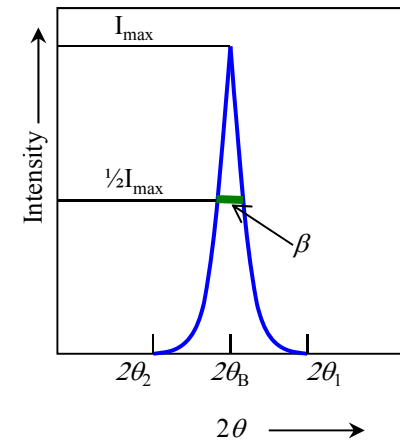
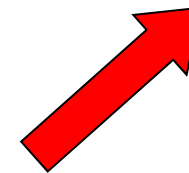
- Scherrer Formula:

$$D = \frac{k\lambda}{\beta \cos \theta} \iff \beta = \frac{k\lambda}{D \cos \theta}$$

$D$  = grain diameter

$\beta$  = peak width in radians at FWHM after correcting for instrumental broadening

$k$  = 0.9 to 1.0 depending upon grain shape

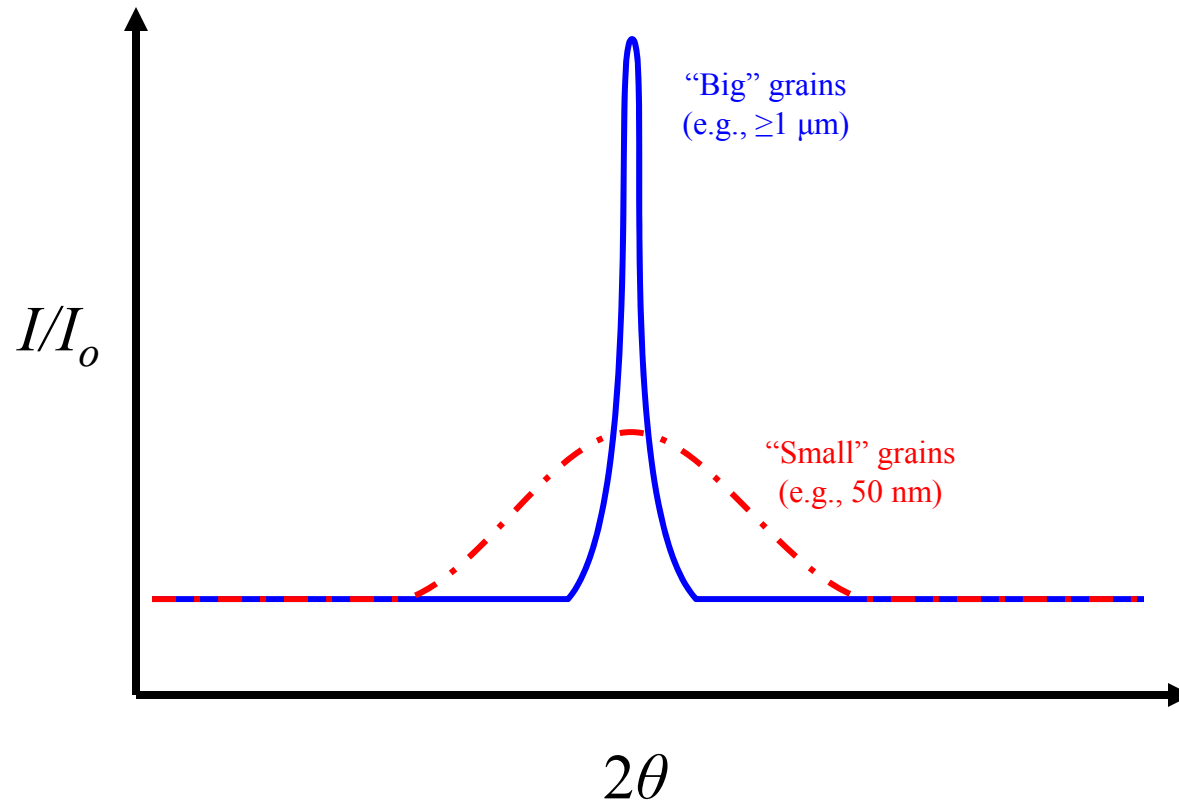


- Equation applies for particle/grain sizes down into the nm range. Works best for particles in the range 2 to 300 nm.

# Scherrer Equation

- This formula was derived based on the assumption of Gaussian peak profiles and small cubic crystals of uniform size (where  $k = 0.94$ ).
- The constant  $k$  actually varies between 0.89 and 1.39 depending upon crystallite size and shape. It is generally assumed to be around 1.0 which provides a precision of  $\pm 10\%$ .
- Be careful because strain also influences peak width.

# Schematic illustration of influence of grain size on peak width



- Keep in mind that peak width can be influenced by strain as well as particle size.

# Broadening due to strain

- Deformation causes local changes in atomic spacing.
- This causes local changes in the  $2\theta$  positions for diffracted x-rays.
- Lattice strain causes peak broadening.
- Broadening due to strain can be represented by the relationship:

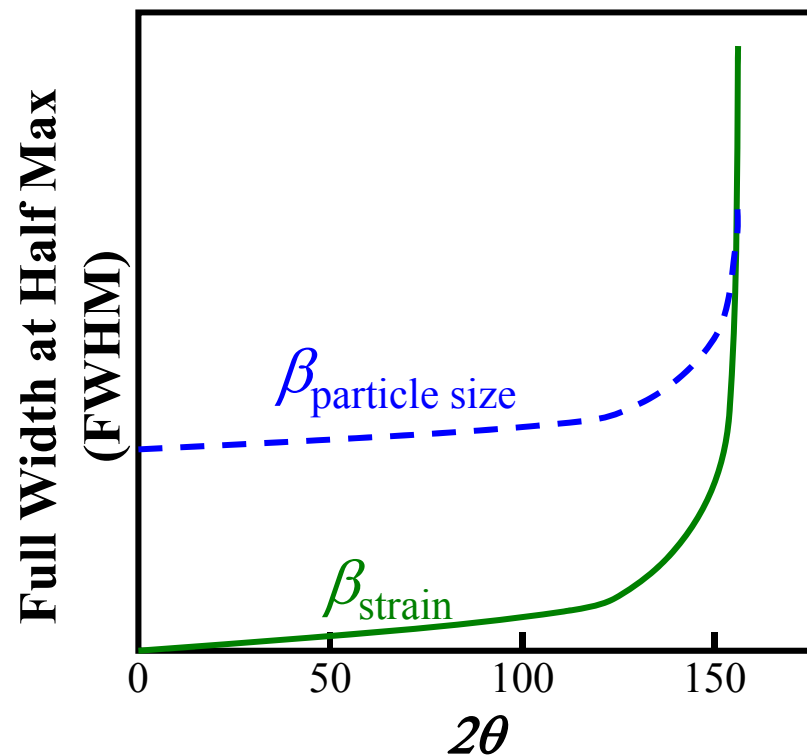
$$\beta_{strain} = \eta \tan \theta$$

Strain in material



## Variation with Bragg Angle

- Amount of peak broadening increases with Bragg angle ( $\theta$ ).
- Most materials exhibit broadening due to crystal size & strain.
- Must use low  $\theta$  peaks to separate effects.



## Now we can do something!

- Width of diffraction peak after subtracting instrumental effects equals the sum of the widths due to lattice strain and crystal size

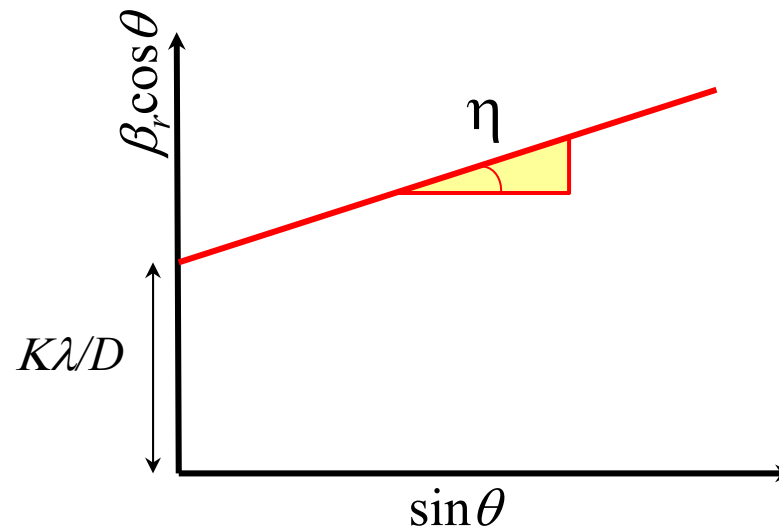
$$\beta_r = \beta_{particle} + \beta_{strain}$$

- Substituting appropriate equations yields:

$$\beta_r = \frac{K\lambda}{D \cos \theta} + \eta \tan \theta \quad \longrightarrow \quad \beta_r \cos \theta = \frac{K\lambda}{D} + \eta \sin \theta$$

This equation is in the general form of a straight line ( $y = mx+b$ ).

- Now, **plot  $\beta_r \cos\theta$  vs.  $\sin\theta$** . Get straight line w/ slope  $\approx$  the lattice strain,  $\eta$ , and an intercept equal to  $K\lambda/D$ .



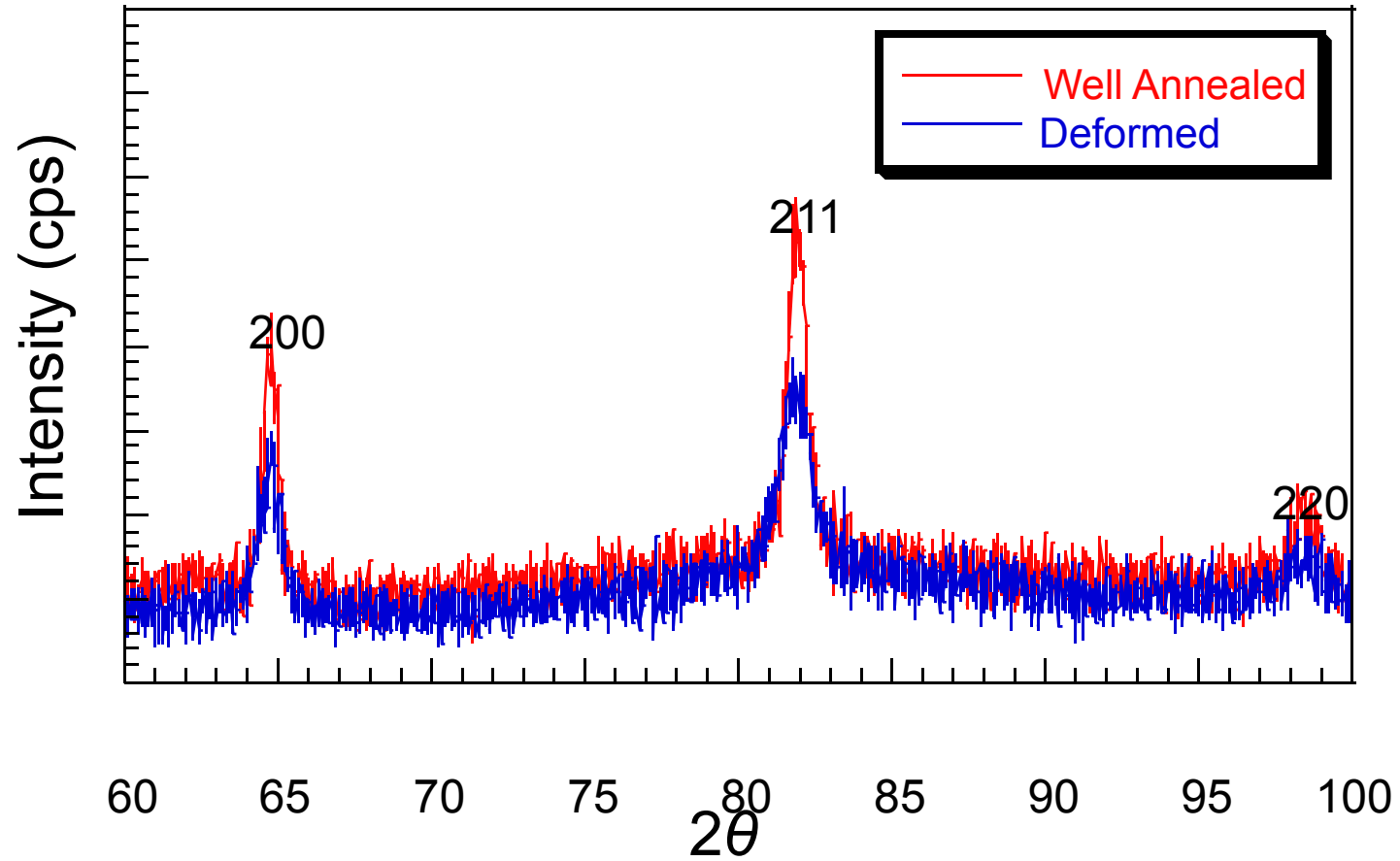
- Basically, the larger the intercept, the smaller the particle size.
- The method works best for smaller crystals.
- Remember, there will be no peak broadening for extremely large crystals.

- Using these relationships you should be able to calculate crystallite size and qualitatively assess strain based upon peak broadening.

# Steps

1. Collect and index diffraction patterns
  - A. Well annealed specimen
  - B. Deformed specimen
2. In well annealed specimen, calculate FWHM for each reflection and determine instrumental broadening to get  $\beta_i$
3. Repeat calculations for experimental (*i.e.*, deformed) specimen to get  $\beta_o$
4. Determine  $\beta_r$  (true width after subtracting instrumental broadening);  
$$\beta_r = \sqrt{\beta_o^2 - \beta_i^2}$$
5. Calculate  $\beta_r \cos\theta$  and plot versus  $\sin \theta$

# Collect and Index



# Steps



Collect and index diffraction patterns

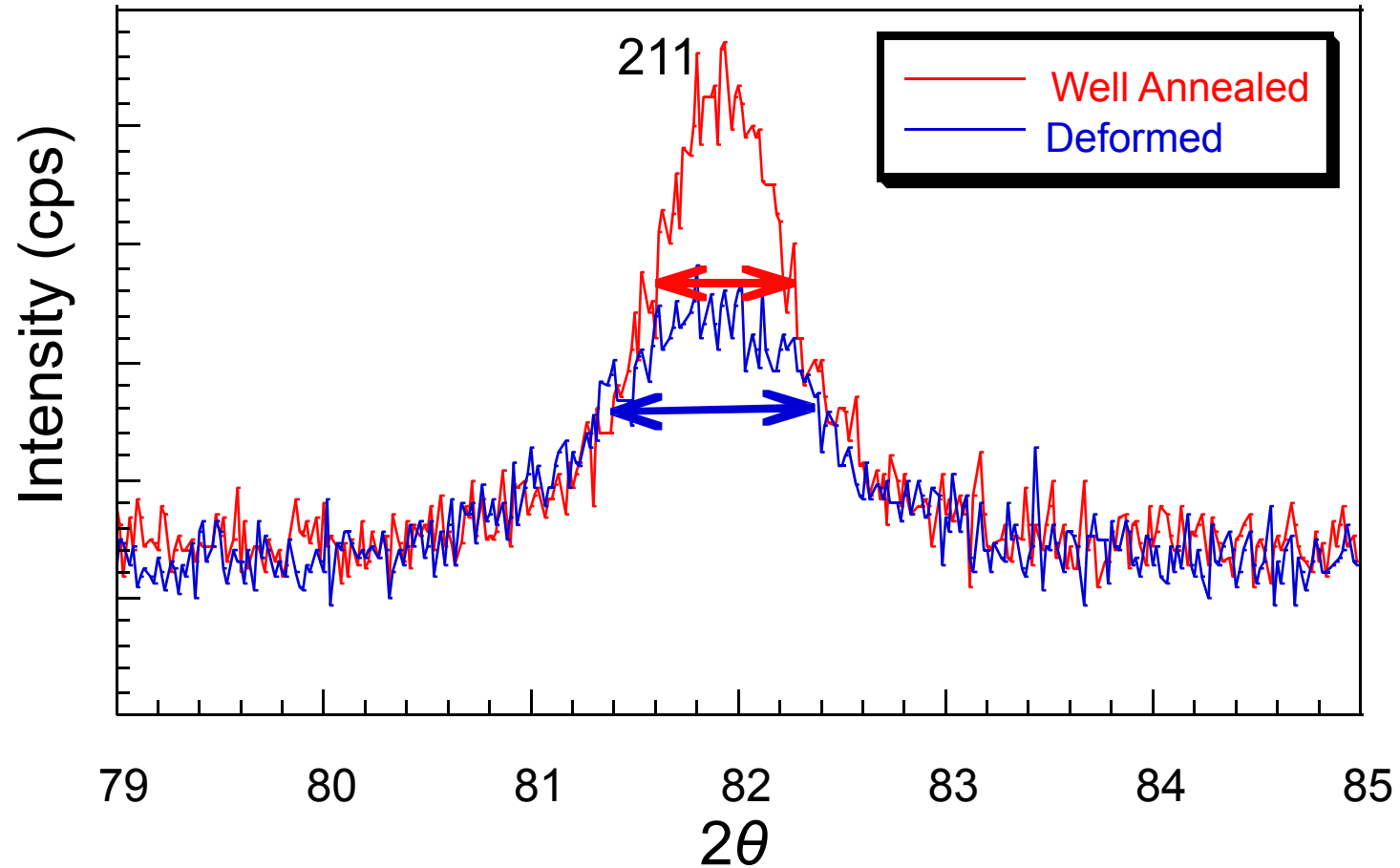
- A. Well annealed specimen
- B. Deformed specimen

- 2. In well annealed specimen, determine FWHM for each reflection and the instrumental broadening to get  $\beta_i$ ;
- 3. Repeat calculations for the deformed specimen to get  $\beta_o$ ;
4. Determine  $\beta_r$  (true width after subtracting instrumental broadening);
5. Calculate  $\beta_r \cos \theta$  and plot versus  $\sin \theta$ .



$$\beta_r = \sqrt{\beta_o^2 - \beta_i^2}$$

# Determine FWHM for Each



Compile Data



# Data

**CuK $\alpha$**   
radiation: 1.54056 Å

**Annealed particles**

Peak#	2 $\theta$ (°)	<i>hkl</i>	FWHM (°)	FWHM (radians) = $\beta_i$
1	64.9	200	0.5	8.73E-03
2	82.0	211	0.6	1.05E-02
3	98.5	220	1.1	1.92E-02

**Deformed particles**                      **CuK $\alpha$**   
radiation: 1.54056 Å

Peak#	2 $\theta$ (°)	sin $\theta$	<i>hkl</i>	FWHM (°) = $\beta_o$	FWHM (radians) = $\beta_o$	$\beta_r = \sqrt{\beta_o^2 - \beta_i^2}$	$\beta_r \cos\theta$
1	64.9	0.5366	200	0.75	1.31E-02		
2	82.0	0.6561	211	1	1.75E-02		
3	98.5	0.7576	220	1.5	2.62E-02		

*Remember to convert all peak widths to radians!*

# Steps



Collect and index diffraction patterns

- A. Well annealed specimen
- B. Deformed specimen



In well annealed specimen, determine FWHM for each reflection and the instrumental broadening to get  $\beta_i$



Repeat calculations for the deformed specimen to get  $\beta_o$

→ 4. Determine  $\beta_r$  (true width after subtracting instrumental broadening;

→ 5. Calculate  $\beta_r \cos \theta$  and plot versus  $\sin \theta$



# Data

**CuK $\alpha$**   
**radiation: 1.54056 Å**

**Annealed particles**

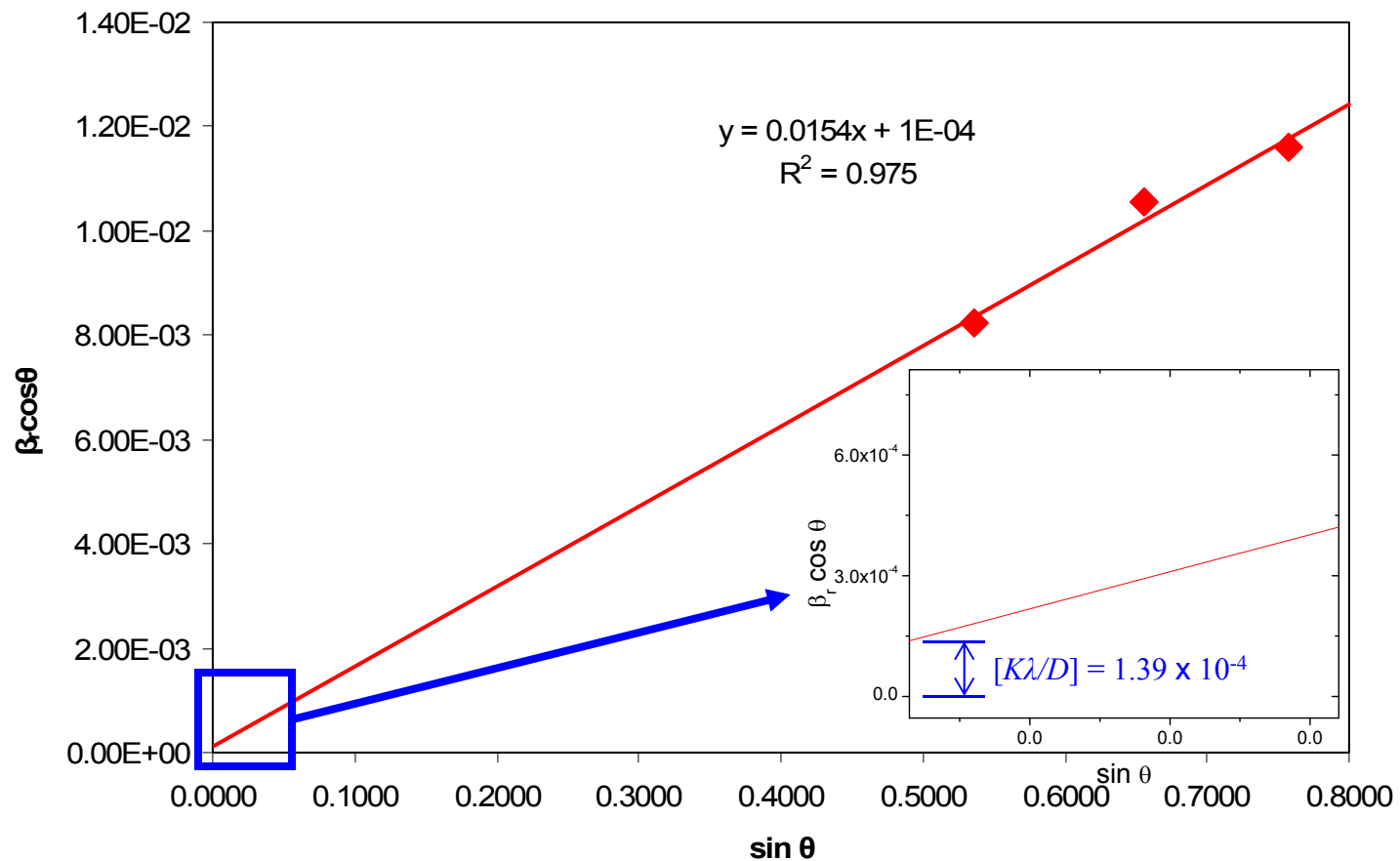
Peak#	2 $\theta$ (°)	<i>hkl</i>	FWHM (°)	FWHM (radians) = $\beta_i$
1	64.9	200	0.5	8.73E-03
2	82.0	211	0.6	1.05E-02
3	98.5	220	1.1	1.92E-02

**Deformed particles**

**CuK $\alpha$  1.54056 Å**

Peak#	2 $\theta$ (°)	sin $\theta$	<i>hkl</i>	FWHM (°) = $\beta_o$	FWHM (radians) = $\beta_o$	$\beta_r = \sqrt{\beta_o^2 - \beta_i^2}$	$\beta_r \cos\theta$
1	64.9	0.5366	200	0.75	1.31E-02	9.76E-03	8.23E-03
2	82.0	0.6561	211	1.0	1.75E-02	1.40E-02	1.05E-02
3	98.5	0.7576	220	1.5	2.62E-02	1.78E-02	1.16E-02

*Plot these two quantities*



- Slope of line is proportional to strain.

$$\beta_r \cos \theta = \frac{K\lambda}{D} + \eta \sin \theta$$

- Y-intercept is related to grain/particle size.

## Particle Size

$$\frac{k\lambda}{D} = 1.39 \times 10^{-4} = \frac{(\sim 1)(1.54056)}{D}$$

$\therefore$

$$D = \frac{1.54056}{1.39 \times 10^{-4}} = 11,083 \text{ \AA} \text{ or } 1.1 \mu\text{m}$$

## Strain

Slope of line = 0.0154 or 1.5%

This method only provides “ballpark” numbers



**Let's stop for now.**

**The rest of this packet includes information on residual stress determination.**

**We'll go through later it if we have time.**